

Article

Osteogenic-Inducing Apatite/Agarose/Gelatin Hybrid Scaffolds Embedding Gold Nanoparticles

María Victoria Cabañas^{1,2}, Paola S. Padilla¹, Mónica Cicuéndez^{1,3,4}, Sandra Sánchez-Salcedo^{1,2,4} ,
Jesús Román^{1,2,*}  and Juan Peña^{1,2,*}

¹ Departamento de Química en Ciencias Farmacéuticas, Facultad de Farmacia, Universidad Complutense de Madrid, UCM, 28040 Madrid, Spain; vcabanas@ucm.es (M.V.C.); paolapadillava@uide.edu.ec (P.S.P.); mcicuend@ucm.es (M.C.); sansanch@ucm.es (S.S.-S.)

² Instituto de Investigación Hospital 12 de Octubre, i+12, 28040 Madrid, Spain

³ Instituto de Investigación Sanitaria del Hospital Clínico San Carlos (IdISSC), 28040 Madrid, Spain

⁴ CIBER de Bioingeniería, Biomateriales y Nanomedicina (CIBER-BBN), 28040 Madrid, Spain

* Correspondence: jeromzar@ucm.es (J.R.); juanpena@ucm.es (J.P.)

Abstract

Objectives: To prepare porous scaffolds combining hydrogel and hydroxycarbonateapatite, enriched with a promising therapeutic agent, gold nanoparticles, to improve bone regeneration. The fabrication procedure is conducted under mild conditions, without toxic or aggressive chemicals, at physiological pH, and low temperatures; **Methods:** Gold nanoparticles (15–20 nm), were obtained by the Turkevith method. The scaffolds were fabricated by the GELPOR3D method, which has demonstrated its ability to integrate thermal labile molecules, during the scaffold fabrication process. The role of these nanoparticles in promoting cell adhesion, proliferation, and mineralization processes in vitro has been studied using osteoprogenitor MC3T3-E1 cells; **Results:** The scaffold fabrication conditions, combined with the surface functionalization of the gold nanoparticles with poly(ethylene glycol), ensure their uniform distribution throughout the scaffold and facilitate their gradual release over 48 h in a physiological medium. A significant increase in the mean cell area and a significant decrease in the circularity index during the early stages of osteoblast differentiation are observed. These pieces of evidence suggest that adequate cell spreading could lead to enhanced proliferation and matrix deposition activity; **Conclusions:** Scaffolds containing these gold nanoparticles exhibited a marked improvement in adhesion, proliferation, and mineralization of preosteoblasts (MC3T3 cells) at the concentrations studied. The functionalization of the nanoparticles, along with the shaping procedure employed, is critical for their homogeneous dispersion throughout the scaffold and their progressive release. The findings confirm the crucial role of gold nanoparticles in the early stages of osteoblast differentiation, which is essential for the transition from premature osteoblasts to mature osteoblasts.

Keywords: functionalized (PEGlyated) gold nanoparticles; bone tissue engineering scaffolds; hydrogels; mineralization; circularity index



Academic Editor: Ruggero Bettini

Received: 21 July 2025

Revised: 15 August 2025

Accepted: 20 August 2025

Published: 25 August 2025

Citation: Cabañas, M.V.; Padilla, P.S.; Cicuéndez, M.; Sánchez-Salcedo, S.; Román, J.; Peña, J. Osteogenic-Inducing Apatite/Agarose/Gelatin Hybrid Scaffolds Embedding Gold Nanoparticles. *Pharmaceutics* **2025**, *17*, 1103. <https://doi.org/10.3390/pharmaceutics17091103>

Copyright: © 2025 by the authors.

Licensee MDPI, Basel, Switzerland.

This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

1. Introduction

The use of porous scaffolds as three-dimensional networks for cell adhesion is a promising strategy for bone tissue regeneration. Ideal scaffolds for bone tissue engineering should be biocompatible, biodegradable, and possess a porous architecture with mechanical properties similar to those of natural bone. Depending on their composition, these

scaffolds can also stimulate osseointegration and osseointegration [1–3]. However, no single material meets all these requirements, leading researchers to explore composite scaffolds that leverage the strengths of various materials. Scaffolds that combine biopolymers and bioactive ceramics are considered particularly promising alternatives [4–6]. Additionally, incorporating a drug or bioactive molecule whose controlled release facilitates the scaffold's integration and performance has become a milestone in the design and fabrication of this type of construct.

Hydrogels, especially those based on natural polysaccharides, such as agarose, have garnered attention due to their resemblance to the extracellular matrix, excellent biocompatibility, and thermosensitive properties [7–10]. Furthermore, the incorporation of another type of hydrogel, gelatin—a type I collagen derivative—enhances the bioactivity of agarose-based scaffolds by introducing cell recognition motifs that facilitate tissue integration [11,12]. Adding calcium phosphates, such as hydroxyapatite, further improves the mechanical performance and bioactivity of hydrogel-based scaffolds, promoting osteoinduction and osteoconduction. Notably, carbonate-substituted hydroxyapatite displays superior bioactive properties compared to stoichiometric hydroxyapatite [13–15].

Scaffold architectures that mimic the hierarchical, interconnected porous structure of bone are essential for bone cell development. Techniques, such as the GELPOR 3D method [16,17], enable the fabrication of scaffolds with customized porosity and the incorporation of thermally sensitive substances, including proteins, nanoparticles, and drugs. This method's advantages—simplicity and low cost—are complemented by the ability to incorporate these substances under physiological conditions, such as low temperatures and neutral pH, without toxic or aggressive solvents. This simultaneous incorporation ensures a homogeneous distribution and integration, resulting in a more controlled and progressive release [18–20]. These scaffolds demonstrate excellent surgical handling, mechanical performance, integration, and gradual degradation when implanted subcutaneously in rats [21], showing no premature decomposition that could compromise their performance [12,22].

Bone morphogenetic protein-immobilized scaffolds have been widely used in bone tissue engineering [23,24], but their clinical application has been limited due to challenges such as low yields, high costs, unwanted ectopic bone formation, and local inflammatory reactions. These facts have prompted the search for alternative strategies that offer improved cost-effectiveness and functional stability for osteoinductive agent-immobilized bone substrates.

An alternative strategy is to incorporate nanomaterials, such as liposomes, metal, ceramic, or polymeric nanoparticles, into the scaffolds [25–28]. In this regard, gold nanoparticles (GNPs) have been widely studied for drug release or tissue engineering, more specifically for bone regeneration [29–35]. Several studies have reported that GNPs can stimulate various biochemical signaling pathways, including p38 MAPK, autophagy, and the Wnt/ β -Catenin Pathway [36,37]. These nanoparticles enhance osteogenic gene expression, alkaline phosphatase activity, and mineralized nodule formation, with both in vitro and in vivo studies demonstrating rapid and effective bone regeneration [35,38,39].

In this study, we present a novel composite scaffold that combines agarose, gelatin, and carbonate-substituted hydroxyapatite, enriched with gold nanoparticles. Our goal was to develop cost-effective, stable, and osteoinductive 3D scaffolds that do not utilize toxic chemicals or solvents. We evaluated the release of gold nanoparticles under physiological conditions and their role in promoting cell adhesion, proliferation, and mineralization processes in vitro, highlighting their potential to advance bone tissue engineering.

2. Materials and Methods

2.1. Synthesis and Characterization of Gold Nanoparticles

Gold nanoparticles (GNPs) were synthesized by reducing tetrachloroauric(III) acid trihydrate with a trisodium citrate dihydrate solution using the method described by Turkevith [40,41]. Briefly, a 50 mL solution of 0.5 mM HAuCl₄ (Sigma–Aldrich, Steinheim, Germany) was prepared and brought to a boil with vigorous stirring. Once the temperature stabilized, 1 mL of a 2% aqueous trisodium citrate solution (Sigma–Aldrich, Steinheim, Germany) was added. The mixture was stirred for 15 min until the solution exhibited a ruby-red coloration. After the color change, the solution was removed from the heat source and allowed to cool while stirring for 24 h. The resulting GNPs suspension was then centrifuged at 14,000 rpm at 4 °C, washed twice with distilled water, suspended in Phosphate Buffered Solution (PBS), and stored at 4 °C for later use.

PEGylated GNPs (GNPs-PEG) were created by conjugating the GNPs with poly(ethylene glycol) methyl ether thiol, PEG-SH (Ernst-Simon-Strasse, Tuebingen, Germany, Mw = 1218.1037). This functionalization was conducted at room temperature by adding 5 mL of PEG-SH solution (7.5 mg/mL) to 10 mL of the GNPs suspension and stirring for 30 min. The resulting GNPs-PEG suspension was centrifuged at 14,000 rpm at 4 °C, washed twice with distilled water, resuspended in 5 mL of PBS, and stored at 4 °C for later use.

The suspensions of both GNPs and functionalized GNPs-PEG were characterized using UV–visible spectrophotometry (UV–vis) in the range of 400–800 nm with a Synergy™ 4 plate reader (BioTek Instrument, Winooski, VT, USA). Their morphology and size were analyzed using Transmission Electron Microscopy (TEM) on a JEOL 1200EX. Hydrodynamic size (DLS) and zeta potential were measured in distilled water using a Malvern Zetasizer Nano ZS90 (Malvern Instruments, Malvern, UK). The concentration of gold nanoparticles in the suspension was determined through inductively coupled plasma-optical emission spectrometry (ICP-OES) using a PerkinElmer Optima 2100 DV ICP apparatus (Waltham, MA, USA). Prior to the analysis, samples were digested overnight at 90 °C in a mixture of HNO₃ and HCl (1:3) and subsequently diluted with distilled water.

2.2. Scaffolds Manufacturing and Characterization

The scaffolds were fabricated using the GELPOR3D method [16,22] as described in Figure 1. For this purpose, agarose powder (Sigma–Aldrich, Steinheim, Germany, for routine use) was suspended in PBS solution (3.5% *w/v*) at pH 7.4 and heated to 90 °C with continuous stirring until a translucent suspension was achieved. The temperature was then reduced to 40 °C before adding gelatin type A (Sigma-Aldrich), ceramic hydroxycarbonateapatite (HCA, synthesized in our laboratory [42]), and the gold nanoparticle suspension. Once the slurry was homogeneous, it was poured into a mold and allowed to gel at room temperature for approximately 5 min. Subsequently, the filaments were withdrawn and the resulting 3D interconnected porous scaffold removed [43] (Figure 1).

The scaffold composition was based on previous results and consisted of (% by weight): 45% agarose, 10% gelatin, and 45% hydroxycarbonateapatite. Two different volumes (1 or 4 mL) of an aqueous gold NP suspension (7.8 µg/mL) were incorporated into the agarose mixture using a syringe: AGH-1Au and AGH-4Au. Additionally, a sample without gold nanoparticles (AGH) was prepared. Scaffolds were also produced without the ceramic component to aid in characterizing the inclusion of gold nanoparticles (AG, AG-1Au, and AG-4Au).

The obtained scaffolds are hydrogels with a high water content (>90%, as determined by Thermogravimetric Analysis) and were used fresh for the various tests conducted. For structural characterization, these scaffolds were freeze-dried and analyzed using Scanning Electron Microscopy (SEM) on a JEOL JSM7600F equipped with an Energy-Dispersive

X-ray Analysis (EDS) detector, X-Ray Diffraction (XRD) on a Philips X'Pert Plus, Cu K α diffractometer, and Fourier Transform Infrared Spectroscopy (FTIR) on a Thermo Nicolet Nexus 470 spectrometer.

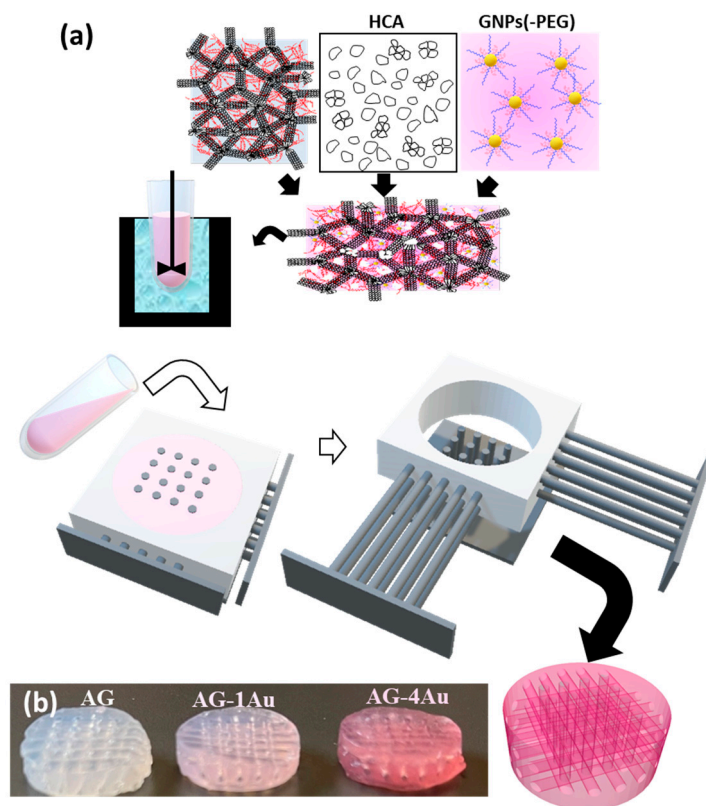


Figure 1. (a) Scaffold fabrication method: GELPOR3D; (b) HCA-free scaffolds with increasing gold nanoparticle content.

2.3. In Vitro Studies

2.3.1. Gold Nanoparticle Release Assays

The release study of GNPs-PEG from the scaffolds was conducted under physiological conditions (pH = 7.4 and 37 °C) using a PBS solution. In each of the triplicate experiments, a scaffold (12 × 2 mm) was immersed in 2 mL of PBS in 12-well plates under mechanical oscillation. The concentration of released gold NPs over time was determined by UV–vis spectrophotometry, measuring the absorbance of the supernatant at $\lambda = 520$ nm. A calibration curve ranging from 0.1 to 4 $\mu\text{g}/\text{mL}$ was created beforehand to determine the concentration of gold nanoparticles.

2.3.2. Cell Assays

Culture of MC3T3-E1 Preosteoblasts

MC3T3-E1 cells were cultured in alpha-MEM (Gibco, Thermo Fisher Scientific, Wilmington, DE, USA) supplemented with 10% fetal bovine serum (FBS) (Gibco), 1 mM L-glutamine (Gibco), and 1% penicillin/streptomycin (Gibco) at 37 °C under a 5% CO₂ atmosphere. Cells were routinely subcultured every 2–3 days, when they reached 80% confluency, using a trypsinization method.

Cell Morphology and Adhesion Assay

Before performing the cell adhesion assay, scaffolds were sterilized under ultraviolet radiation for 1 h. MC3T3-E1 cells were then seeded onto the surface of the hydrogels at a density of 2×10^4 cells per scaffold (in 12-well plates) and incubated for 9 days in culture

medium without any specific differentiation factors. Phalloidin-rhodamine conjugate (red) was used to stain the F-actin filaments of the cytoskeleton, while DAPI (4',6-diamidino-2-phenylindole, a blue fluorescent marker; Fluoroshield (Sigma-Aldrich, St. Louis, MO, USA) was used to stain the cell nuclei. After removing the culture medium and washing the cells twice with PBS, 0.5 mL of a 4% paraformaldehyde (PFA) solution (Sigma-Aldrich) was added per well to fix the cells, followed by incubation in an oven at 37 °C for 20 min. The PFA was then thoroughly washed out with PBS, and 1 mL of PBS was left in each well. After this, 20 µL of phthalocyanine was added and incubated for 20 min with agitation in darkness. DAPI staining was then performed at a dilution of 1:5000 for 10 min. Finally, samples were observed using the Evos FL Cell Imaging System (Advanced Microscopy Group, Bothell, WA, USA). Control scaffolds (AGH without GNPs-PEG) were also included.

For fluorescence microscopy, an EVOS FL Cell Imaging System inverted optical microscope equipped with three types of LED light (IEX (nm); IEM (nm)) from AMG (Advanced Microscopy Group, Bothell, WA, USA) was used. The red channel was used to observe the cytoskeleton labelled with the probe Phalloidin-rhodamine conjugate (540/565) and the blue channel to observe the cell nucleus labelled with DAPI (358; 461).

For quantitative analysis of cell morphology, images were captured from triplicate samples (10 randomly chosen fields of view each) on a given number ($n = 50$) of randomly chosen cells. Cell circularity is used to provide a quantification of cell shape. The formula used for determining the circularity index (CI) is $CI = 4\pi \cdot (\text{Area}/\text{Perimeter}^2)$. A value of 1.0 indicates a perfect circle and a value of 0.0 a totally elongated structure.

Cell Proliferation Assay

Cells were seeded onto the surface of the hydrogels at a density of 2×10^4 cells per hydrogel (in 12-well plates) and incubated for 1, 6, and 8 days in culture medium without any specific differentiation factors. Cell proliferation was assessed using the Alamar Blue test (Invitrogen™ Thermo Fisher Scientific, Waltham, MA, USA), with fluorescence measured at 560–570 nm using the Synergy™ 4 reader (Biotek). Control scaffolds (AGH without GNPs-PEG) were included.

Mineralization Assay

Mineralization was assessed using alizarin red S (ARS) staining, indicated by the formation of calcium nodules in MC3T3-E1 cells. After washing the wells with PBS, the cells were fixed with 70% ethanol for 1 h at 4 °C. Subsequently, the cells were treated with a 40 mM ARS solution (0.136 g/10 mL) at pH 4.2 for 30 min, followed by washing with PBS to remove non-specific staining. To quantify calcium mineralization, the dye was eluted using 10% (*w/v*) cetylpyridinium chloride (Sigma-Aldrich) in 10 mM sodium phosphate buffer at pH 7 for 30 min, and the absorbance was measured at 620 nm using the Synergy™ 4 reader (Biotek Instrument). Control scaffolds (AGH without GNPs-PEG) were also included.

Statistical Analysis

Data are expressed as means \pm standard deviations from three independent experiments, each carried out in triplicate. Statistical analysis was performed using the Statistical Package for the Social Sciences (SPSS) software, version 22. Comparisons among groups were made using analysis of variance (ANOVA), with the Scheffé test employed for post-hoc evaluations. In all the statistical evaluations, a *p*-value of <0.05 was considered statistically significant.

3. Results and Discussion

3.1. Gold Nanoparticles

The ruby-red color of the gold nanoparticle (GNP) suspension (Figure 2) provides critical qualitative information. A purple coloration indicates the aggregation of the gold nanoparticles. The UV–vis absorption spectrum of these red GNPs (Figure 2) displays a localized surface plasmon resonance (LSPR) at 520 nm, which is characteristic of spherical GNPs with sizes ranging from 15 to 20 nm [44,45]. Additionally, the sharp absorption band suggests that these GNPs are uniformly distributed and not aggregated.

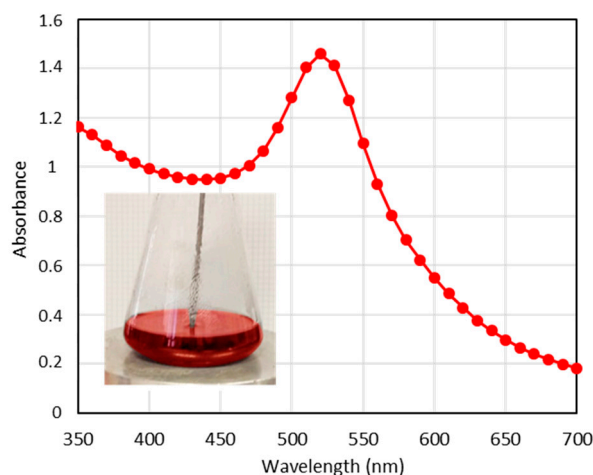


Figure 2. UV–vis spectra corresponding to the GNPs. (inset): Photograph of GNPs suspension during the synthesis process.

To determine the size and shape of the GNPs by TEM, a drop of an aqueous suspension containing the nanoparticles was placed on a copper grid and allowed to evaporate. A TEM image of the GNPs is shown in Figure 3a, revealing a quasi-spherical morphology with a monomodal size distribution. The statistical size was measured to be 14 ± 1.9 nm using ImageJ software version 1.54k, based on measurements of 130 nanoparticles. Additionally, the inset in the figure shows that the average hydrodynamic size obtained from DLS was 15.60 ± 0.3 nm, indicating satisfactory dispersibility [46]. The zeta potential of the prepared GNPs was found to be -15.6 mV, which can be attributed to the negative charge of the citrate groups on the GNPs' surface.

These freshly prepared GNPs exhibit stability; however, aggregation occurs after four days. To enhance their stability, these nanoparticles were functionalized with PEG-SH. The UV–vis spectrum of the modified nanoparticles (GNPs-PEG) also exhibits LSPR at 520 nm, maintaining the spherical shape and size, as indicated in the TEM image in Figure 3b. The GNPs-PEG possess a zeta potential of -20.5 mV, similar to the non-functionalized gold nanoparticles, but with a higher hydrodynamic size of 24.36 ± 3.1 nm due to the conjugation of PEG-SH groups onto the GNPs' surface (Figure 3b).

3.2. Scaffolds Manufacturing and Characterization

Agarose, a thermogelling polysaccharide, can form a gel at room temperature within minutes without requiring chemical crosslinking agents. This biocompatible polysaccharide is ideal for fabricating scaffolds under physiological conditions. However, since agarose lacks cell-recognition motifs, the bioactivity of such composites must be significantly enhanced by incorporating specific proteins. In fact, we recently demonstrated that the addition of gelatin to apatite/agarose scaffolds improves their interaction with New Zealand rabbit MSCs [12].

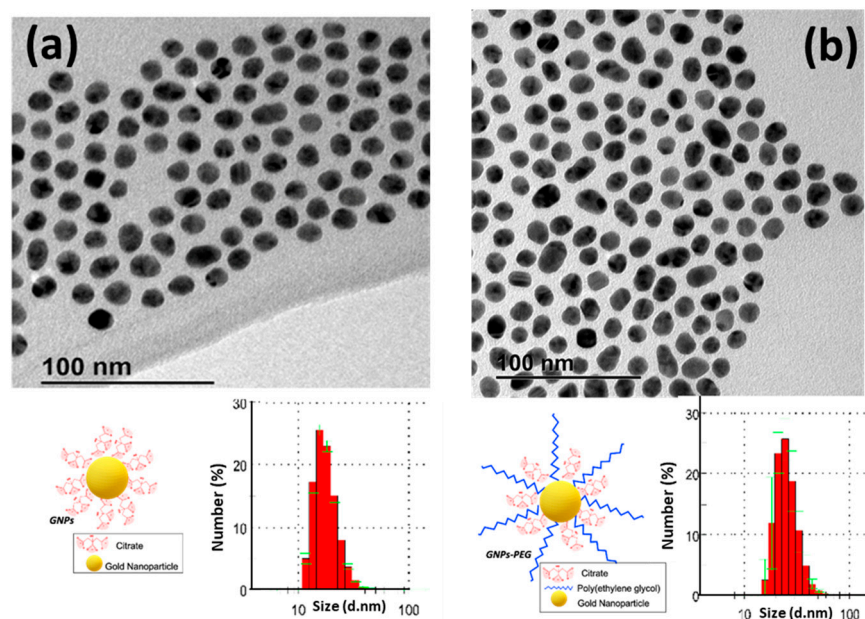


Figure 3. TEM images and size histograms measured by DLS (inset) of the two types of gold nanoparticles synthesized: (a) GNPs; (b) GNPs-PEG. The green lines in the histograms represents the error bars.

Despite the findings by Tentor et al. [47], which indicated that the introduction of GNPs into chitosan/pectin hydrogels affects the gelation temperature, our study found that the incorporation of gold nanoparticles in the amounts analyzed did not affect the agarose gelation process, which is crucial for scaffold fabrication. The presence of these nanoparticles within the scaffold was confirmed by the color change from white to ruby-red (Figure 1b). The homogeneous distribution of the nanoparticles within the scaffold was apparent to the naked eye due to the ruby-red coloration previously observed during the preparation of the nanoparticles. It was essential to cool the agarose suspension to 40 °C; otherwise, gold particle aggregation was observed, evidenced by the color change of the scaffold from ruby-red to purple, which had also been noted during the optimization process of gold nanoparticles.

Purple agarose scaffolds were observed when non-functionalized GNPs were used, indicating that these gold NPs tend to aggregate within the scaffold matrix (Figure 4a). These aggregated gold particles could not be released from the scaffold when immersed in a PBS solution, as shown by the absence of the characteristic LSPR in the UV–vis absorption spectrum of the supernatant collected after 12 h of immersion (Figure 4a).

In contrast, when GNPs-PEG were employed for scaffold fabrication, the aggregation problem was eliminated, resulting in ruby-red-colored scaffolds (Figure 4b). Furthermore, these scaffolds released the functionalized gold nanoparticles upon contact with an aqueous medium, as confirmed by the appearance of an absorption maximum at 520 nm in the UV–vis spectrum of the supernatant (Figure 4b).

Therefore, the functionalization of GNPs with PEG-SH promotes proper and stable incorporation into the scaffolds, allowing for nanoparticle release in a physiological medium. This phenomenon highlights the significance of color variation as a marker of undesirable gold particle aggregation.

The XRD patterns of scaffolds containing gold NPs are similar to those of scaffolds without gold nanoparticles. They exhibit diffraction maxima corresponding to carbonate apatite and an amorphous structure associated with agarose and gelatin [43]. Additionally, as the scaffolds were prepared in PBS, maxima corresponding to NaCl were also detected (see Supporting Information, Figure S1). The presence of GNPs-PEG cannot be identified using

FTIR spectroscopy, as only bands attributed to agarose, gelatin, or hydroxycarbonateapatite (HCA) are observable (see Supporting Information, Figure S2).

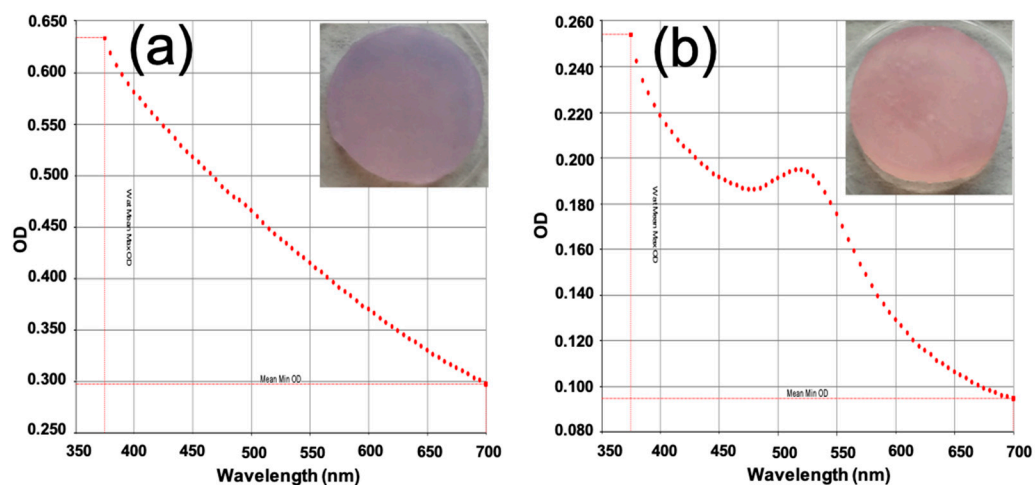


Figure 4. UV-vis absorption spectra of the supernatant solution remaining after 12 h of immersion in PBS of HCA-free scaffolds containing (a) GNPs or (b) GNPs-PEG.

Regarding the characterization of the scaffold morphology, Figure 5 presents photographs of scaffolds fabricated with and without ceramic components. The scaffolds exhibit a homogeneous coloration that intensifies with increasing GNPs-PEG concentration, suggesting an even distribution of nanoparticles within the hybrid matrix. This figure also includes magnified images of the scaffold interior, revealed by cutting the surface, which allows for the quantification of the 900 μm channels, their orientation in three spatial dimensions, and their connectivity.

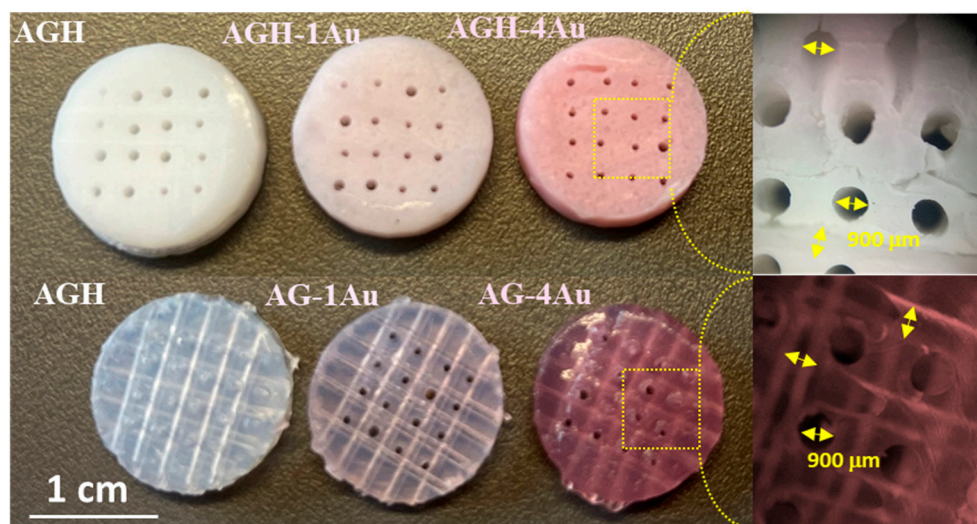


Figure 5. Photographs of designed porous scaffolds fabricated with varying GNPs-PEG content: without (**bottom image**) and with ceramic (**top image**).

Scanning Electron Microscopy (SEM) further enhances the morphological characterization of the scaffolds at much higher magnifications. Figure 6 shows the complete integration of ceramic particles within the agarose/gelatin matrix. Specifically, Figure 6a depicts the characteristic honeycomb morphology of the lyophilized gels, resulting from the sublimation of ice crystals during the freeze-drying process. As discussed in our previous publications [19,43], lyophilization not only serves as a preservation method but also creates additional porosity, resulting in 100–200 μm parallel pores that contribute to better scaffold inte-

gration and performance. Figure 6b confirms that the ceramic particles are fully embedded within the agarose/gelatin matrix, which forms the walls of the honeycomb cells. Backscattered electron SEM images at varying magnifications (Figure 6c,d) reveal the presence of gold nanoparticles (indicated by red arrowheads), measuring approximately 15–20 nm, dispersed throughout the scaffold and embedded within the agarose/gelatin matrix.

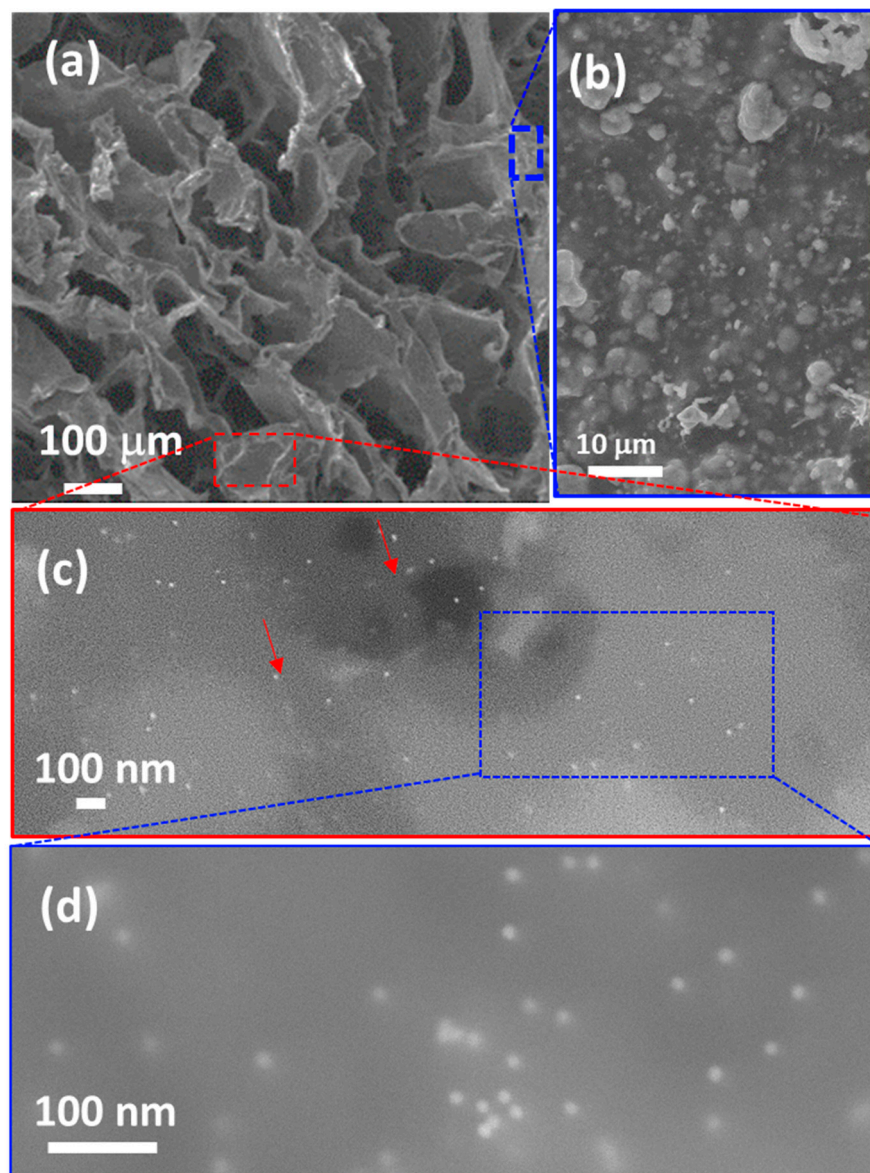


Figure 6. (a,b) SEM images and (c,d) backscattered electron SEM images of the AGH-4Au scaffold cross-section. Red arrows point gold nanoparticles.

3.3. Scaffolds In Vitro Behavior

The rate at which nanoparticles are released from the scaffold is a crucial factor in assessing their behavior once implanted. Release studies indicate that the GNPs-PEG are slowly released from the AGH-xAu scaffolds when submerged in a physiological solution. The characteristic maximum at 520 nm, associated with plasmon resonance, is observed in the UV-vis spectra of the supernatant resulting from the scaffold immersion (Figure 7a). As expected, the AGH-4Au scaffolds, which contain the highest amount of gold nanoparticles, release larger quantities of these particles (Figure 7b). Regardless of the specific scaffold, both types exhibit a similar release pattern, with around 80% of GNPs-PEG released in the initial 24 h and complete liberation by 48 h.

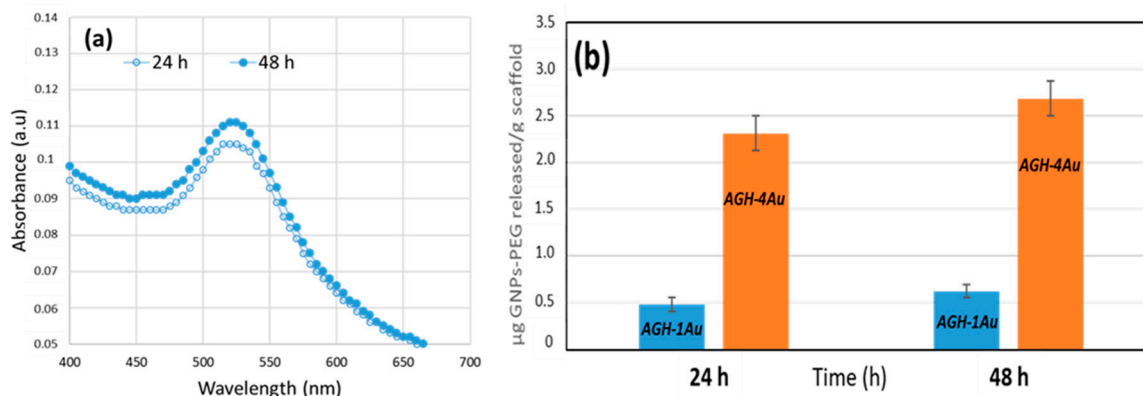


Figure 7. Release of GNPs-PEG from the scaffolds in PBS at 37 °C as a function of time. (a) UV-vis spectra of the supernatant at various time points for the AGH-1Au scaffold immersion; (b) cumulative amount released as a function of time from different scaffolds.

In examining the cell response to the prepared scaffolds, it is important to consider that the differentiation process of mouse osteoprogenitor MC3T3-E1 cells into osteoblasts, under certain stimuli, occurs in three stages: (a) cell proliferation, (b) matrix maturation, and (c) matrix mineralization. During the early stage (days 1–9), MC3T3-E1 cells replicate actively but do not express alkaline phosphatase (ALP) or form mineralized nodules. In the second stage (days 9–16), the cells begin to produce ALP. By the third phase (after day 16), the cells demonstrate both ALP activity and the formation of mineralized nodules [48]. Matrix maturation and mineralization are typically enhanced by allowing the cells to reach confluence and by adding specific osteogenic factors. Therefore, the early stage is crucial for the transition of premature osteoblasts into mature osteoblasts. [49,50]. In this context, cell adhesion to 3D hydrogels plays a critical role in enabling subsequent cellular processes [51]. Good cell adhesion is facilitated when the surface topography offers multiple binding points, thereby increasing the interface area between the surface and cells [52,53]. Enhanced cell spreading facilitates osteoblast differentiation in osteoprogenitor cells and promotes matrix deposition during bone remodeling [54]. It has been shown that cell shapes influence whether human mesenchymal stem cells commit to osteogenic or adipogenic lineages [55]. Since cell shapes relate to the expression of cytoskeletal proteins and integrins, proper regulation of the cytoskeleton may be essential for the osteoblast differentiation process [56].

In this study, we performed a morphological analysis of MC3T3-E1 cells, as the *in vitro* shape of the cells significantly impacts the outcome of osteoblast differentiation. Figure 8a presents representative images of the morphology of MC3T3-E1 preosteoblasts adhered to the surfaces of AGH and AGH-4Au scaffolds after 9 days of culture in cell medium without specific differentiation factors. These conditions serve as a control (AGH, without gold nanoparticles) and reflect the maximum concentration of nanoparticles present in the hydrogel AGH-4Au (2.8 µg/g scaffold). In the images, the cell nucleus is stained blue, while F-actin filaments are labelled in red. The cells appear to be firmly anchored to the surfaces of both scaffolds via filopodia, and there is evidence of intercellular communication. Filopodia are thin, finger-like projections of the cell membrane that extend to explore the cellular environment. They bind to extracellular structures, forming adhesion complexes known as focal adhesions. Quantification of these morphological changes was based on cell shape, as measured by the circularity index, and cell spreading, as determined by the mean cell area. [53]. The results in Figure 8b indicate that the circularity index of preosteoblasts cultured on hydrogels loaded with gold nanoparticles (AGH-1Au and AGH-4Au) significantly decreased (** $p < 0.05$) compared to those cultured on hydrogels without nanoparticles (AGH). Conversely, Figure 8c shows a significant increase (* $p < 0.05$) in the mean cell area of preosteoblasts seeded on AGH-4Au hydrogels, compared to those cultured

on AGH hydrogels without gold nanoparticles. There were no significant differences in the mean cell area between preosteoblasts seeded onto AGH hydrogels and those seeded onto hydrogels with a lower concentration of nanoparticles (AGH-1Au, 0.7 $\mu\text{g/g}$ scaffold). The cells on AGH-4Au scaffolds exhibited both a significantly increased mean cell area and a significantly decreased circularity index during the early stage of osteoblast differentiation (9 days). These results suggest adequate cell spreading, which may lead to enhanced proliferation and matrix deposition activity, as will be illustrated in Figure 9.

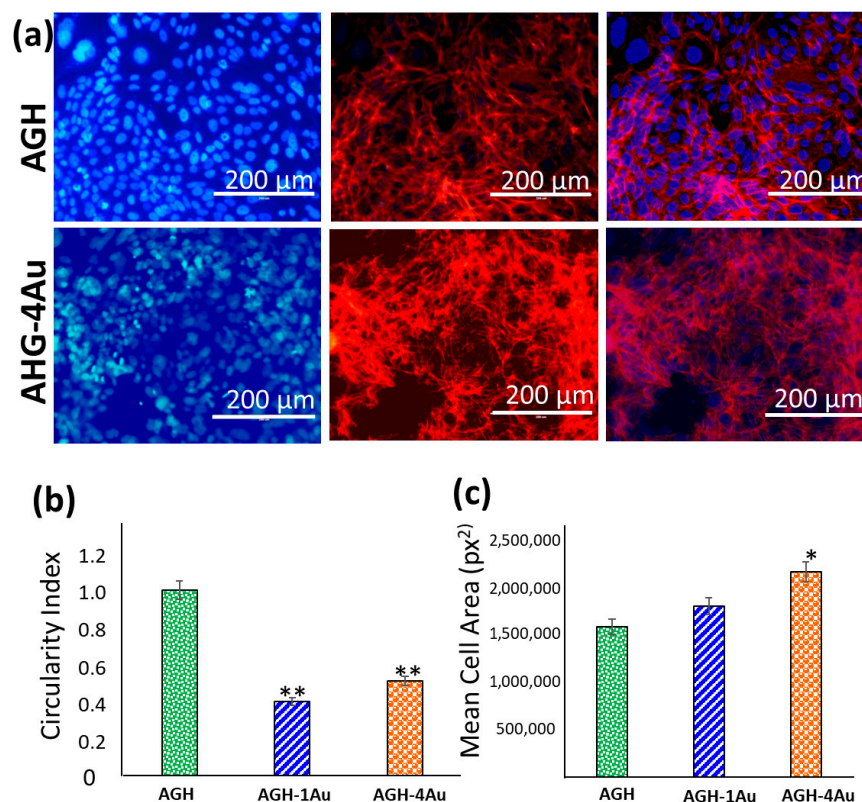


Figure 8. Cell adhesion and morphology of MC3T3-E1 preosteoblasts seeded onto AGH (without NPs) and AGH-4Au hydrogels after 9 days of culture. (a) Representative images obtained by fluorescence laser scanning microscopy. Atto 565-phalloidin and DAPI were used as fluorescence probes to stain the F-actin microfilaments and cell nucleus, respectively. Quantification of cell morphology in terms of (b) circularity index and (c) mean cell area (pixel²) of MC3T3-E1 preosteoblasts cultured onto AGH, AGH-1Au, and AGH-4Au hydrogels. Error bars are standard error of the mean, where $n = 50$. * $p < 0.05$ and ** $p < 0.01$, statistical differences *vs* AGH condition.

Once mineralization is complete, osteoblasts produce vast extracellular calcium deposits, indicating successful *in vitro* bone formation. These deposits can be specifically stained bright orange-red using Alizarin Red S. Detection of functional mineralization is commonly used to characterize osteoblasts *in vitro*. For this reason, the *in vitro* mineralization process of osteoblasts has been extensively employed as a model for testing the effects of drug treatments and mechanical loading on bone cell differentiation and bone formation [57,58].

In this study, we evaluated the cellular differentiation of preosteoblasts (MC3T3-E1) seeded directly onto the surface of scaffolds: one without any gold nanoparticles (AGH) and two with different concentrations of gold nanoparticles (AGH-1Au and AGH-4Au). Our focus was on the early and late phases of the differentiation process, as these are critical stages. Cell proliferation, representing the early stage, was assessed after 1, 6, and 8 days of culture in basal medium without differentiation factors. Meanwhile, cellular

mineralization was evaluated after 15 days under the same experimental conditions. The results, shown in Figure 9a, indicate that preosteoblasts seeded onto the hydrogels, whether in the presence (AGH-1Au and AGH-4Au) or absence (AGH) of gold nanoparticles, were capable of proliferating over time (from days 1 to 8). These facts demonstrate that the final composition of the hydrogels, in terms of the agarose-gelatin-hydroxyapatite ratios and the concentrations of gold nanoparticles used in this study, does not exhibit cytotoxicity toward immature preosteoblasts, allowing for their proliferation. Notably, while there were no significant differences in the proliferation of preosteoblasts seeded onto scaffolds without gold nanoparticles (AGH) compared to those on hydrogels with the lowest concentration of nanoparticles (AGH-1Au), significant differences were observed in the proliferation of preosteoblasts on hydrogels with the higher concentration of nanoparticles (AGH-4Au) compared to those with the lower concentration (AGH-1Au) at both 6 (# $p < 0.05$) and 8 days (& $p < 0.05$). Further, significant differences were also noted when comparing cellular proliferation on AGH-4Au scaffolds with the control condition (* $p < 0.05$), which consisted of hydrogels without nanoparticles (AGH). Thus, it can be concluded that hydrogels with the highest concentration of gold nanoparticles (AGH-4Au, 2.8 $\mu\text{g/g}$ hydrogel) allowed for greater cellular proliferation of immature preosteoblasts, which is essential for the ongoing cellular differentiation process.

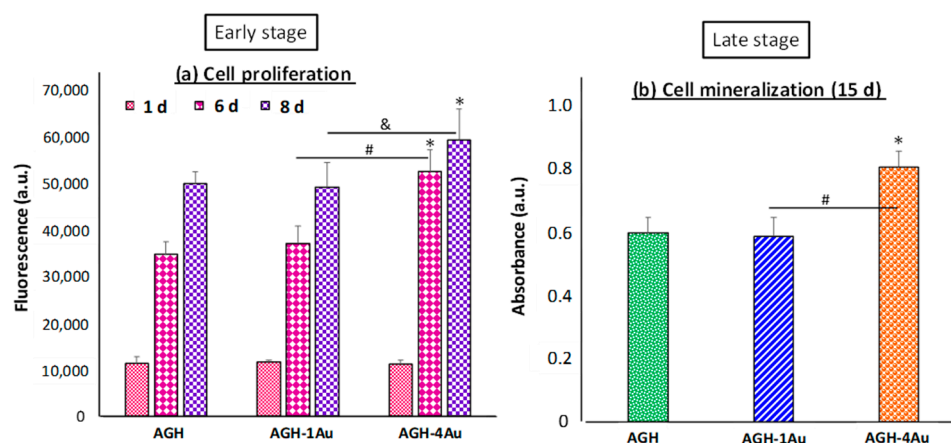


Figure 9. Cell differentiation process of MC3T3-E1 preosteoblasts seeded onto AGH (without NPs) and AGH-1Au and AGH-4Au hydrogels. (a) Cell proliferation (early stage) of preosteoblasts cultured for 1, 6, and 8 days measured by alamar blue test. (b) Cell mineralization (late stage) of preosteoblasts after 15 days of culture evaluated by alizarin red staining. * $p < 0.05$, statistical differences vs. AGH condition. # $p < 0.05$, AGH-1Au vs. AGH-4Au at 6 days in cell proliferation and at 15 days in cell mineralization. & $p < 0.05$ AGH-4Au vs. AGH-1Au after 8 days of proliferation.

In terms of cellular mineralization in the late phase, the results shown in Figure 9b demonstrate significant differences in calcium deposition by preosteoblasts cultured for 15 days on hydrogels with the highest concentration of NPs (AGH-4Au) compared to results obtained from cells cultured on AGH control hydrogels (* $p < 0.05$) and on hydrogels with the lowest concentration of gold nanoparticles (# $p < 0.05$). This study supports Quarles et al.'s classification, indicating that the early stage of osteoblast differentiation is a critical transition from immature to mature osteoblasts. Various studies have indicated that gold nanoparticles ranging from 5 to 20 nm in size enhance osteoblastic proliferation and differentiation, as these nanoparticles interact with the plasma membrane and influence the metabolic MAPK (p38 mitogen-activated protein kinase) pathway regulating osteogenic genes [35,59–61]. The findings regarding cellular responses to the materials underscore the significance of rapid gold NPs release (24–48 h) from the scaffolds (Figure 7),

which promotes the early phase of differentiation from immature preosteoblasts to mature osteoblasts, which is a vital stage that is placed between days 1 and 9.

This study establishes a baseline for incorporating gold nanoparticles into this system and identifies a concentration that yields positive and significant outcomes. Nanoparticles embedded in these hydrogels may serve as an excellent method to enhance the functionality of the composite scaffold, demonstrating therapeutic potential for bone tissue engineering by promoting the viability, proliferation, and mineralization of osteoprogenitor MC3T3 cells.

4. Conclusions

A three-dimensional bone tissue-engineered scaffold with osteogenic properties using only mild aqueous conditions, without any toxic chemicals, was fabricated through the GELPOR3D method. This technique enables the direct incorporation of labile molecules during scaffold fabrication at low temperatures. In this work, we demonstrated that spherical gold nanoparticles (14 nm) can be homogeneously dispersed throughout the scaffolds. To enhance the stability of these gold nanoparticles in suspension and facilitate their inclusion in the scaffolds, we functionalized them with thiolated polyethylene glycol.

The nanoparticles are gradually released into a physiological medium, with complete release occurring after 48 h. Our findings highlight the significant potential of gold nanoparticles as bioactive agents for bone regeneration. Scaffolds containing these nanoparticles exhibited a marked improvement in adhesion, proliferation, and mineralization of preosteoblasts (MC3T3 cells) at the concentrations studied. The results confirm that the early stages of osteoblast differentiation are critical for the transformation from premature to mature osteoblasts. Future studies are needed to evaluate different concentrations of gold nanoparticles in order to determine the optimal dosage that provides significant benefits without compromising cell viability.

Supplementary Materials: The following supporting information can be downloaded at: <https://www.mdpi.com/article/10.3390/pharmaceutics17091103/s1>, Figure S1: XRD patterns corresponding to scaffolds fabricated in PBS. (a) AGH-4Au scaffold prepared with GNPs-PEG; (b) AGH scaffold prepared without NPs; (c) Agarose-4Au scaffold prepared without ceramic (arrows indicate diffraction maxima corresponding to NaCl); Figure S2. FTIR spectra corresponding to AGH-4Au scaffold fabricated in PBS. As well as their components: Agarose, gelatin, and hydroxycarbonateapatite.

Author Contributions: Conceptualization, M.V.C., J.R. and J.P.; methodology, P.S.P., M.V.C. and S.S.-S.; formal analysis, M.V.C., M.C., S.S.-S., J.R. and J.P.; writing—original draft preparation, M.V.C., P.S.P., M.C. and S.S.-S.; writing—review and editing, M.V.C., J.R. and J.P.; visualization, and supervision, M.V.C., M.C., S.S.-S., J.R. and J.P.; funding acquisition, M.V.C., M.C., S.S.-S. and J.P. All authors have read and agreed to the published version of the manuscript.

Funding: The authors acknowledge the funding received from the Carlos III Health Institute through the FIS grant number PI20/01384, which was co-financed with European Regional Development Fund (ERDF) funds from the European Union. The funding obtained from the Regional Government (Comunidad de Madrid, Spain) and the Universidad Complutense de Madrid PR17/24-31900 and PR17/24-31923 (BIO4REG and MacroArt) is also acknowledged.

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not applicable.

Data Availability Statement: All data supporting the findings of this study are included in the present manuscript. Further inquiries can be directed to the corresponding author.

Acknowledgments: The authors want to acknowledge CAI Difraccion de Rayos X (Universidad Complutense de Madrid, Spain) and ICTS Centro Nacional de Microscopia Electrónica (UCM, Madrid, Spain) for their support.

Conflicts of Interest: The authors declare no conflicts of interest.

References

1. Thangavel, M.; Elsen Selvam, R. Review of Physical, Mechanical, and Biological Characteristics of 3D-Printed Bioceramic Scaffolds for Bone Tissue Engineering Applications. *ACS Biomater. Sci. Eng.* **2022**, *8*, 5060–5093. [[CrossRef](#)] [[PubMed](#)]
2. Chen, T.; Jiang, Y.; Huang, J.P.; Wang, J.; Wang, Z.K.; Ding, P.H. Essential Elements for Spatiotemporal Delivery of Growth Factors within Bio-Scaffolds: A Comprehensive Strategy for Enhanced Tissue Regeneration. *J. Control. Release* **2024**, *368*, 97–114. [[CrossRef](#)]
3. Xie, C.; Ye, J.; Liang, R.; Yao, X.; Wu, X.; Koh, Y.; Wei, W.; Zhang, X.; Ouyang, H. Advanced Strategies of Biomimetic Tissue-Engineered Grafts for Bone Regeneration. *Adv. Healthc. Mater.* **2021**, *10*, 2100408. [[CrossRef](#)]
4. Winkler, T.; Sass, F.A.; Duda, G.N.; Schmidt-Bleek, K. A Review of Biomaterials in Bone Defect Healing, Remaining Shortcomings and Future Opportunities for Bone Tissue Engineering: The Unsolved Challenge. *Bone Jt. Res.* **2018**, *7*, 232–243. [[CrossRef](#)] [[PubMed](#)]
5. Abdollahi, F.; Saghatchi, M.; Paryab, A.; Malek Khachatourian, A.; Stephens, E.D.; Toprak, M.S.; Badv, M. Angiogenesis in Bone Tissue Engineering via Ceramic Scaffolds: A Review of Concepts and Recent Advancements. *Biomater. Adv.* **2024**, *159*, 213828. [[CrossRef](#)]
6. dos Santos, V.I.; Chevalier, J.; Fredel, M.C.; Henriques, B.; Gremillard, L. Ceramics and Ceramic Composites for Biomedical Engineering Applications via Direct Ink Writing: Overall Scenario, Advances in the Improvement of Mechanical and Biological Properties and Innovations. *Mater. Sci. Eng. R Rep.* **2024**, *161*, 100841. [[CrossRef](#)]
7. Malda, J.; Visser, J.; Melchels, F.P.; Jüngst, T.; Hennink, W.E.; Dhert, W.J.A.; Groll, J.; Hutmacher, D.W. 25th Anniversary Article: Engineering Hydrogels for Biofabrication. *Adv. Mater.* **2013**, *25*, 5011–5028. [[CrossRef](#)]
8. Dattilo, M.; Patitucci, F.; Prete, S.; Parisi, O.I.; Puoci, F. Polysaccharide-Based Hydrogels and Their Application as Drug Delivery Systems in Cancer Treatment: A Review. *J. Funct. Biomater.* **2023**, *14*, 55. [[CrossRef](#)]
9. Witzler, M.; Büchner, D.; Shoushrah, S.H.; Babczyk, P.; Baranova, J.; Witzleben, S.; Tobiasch, E.; Schulze, M. Polysaccharide-Based Systems for Targeted Stem Cell Differentiation and Bone Regeneration. *Biomolecules* **2019**, *9*, 840. [[CrossRef](#)]
10. Lee, J.H.; Kim, H.W. Emerging Properties of Hydrogels in Tissue Engineering. *J. Tissue Eng.* **2018**, *9*, 2041731418768285. [[CrossRef](#)]
11. Samadian, H.; Khastar, H.; Ehterami, A.; Salehi, M. Bioengineered 3D Nanocomposite Based on Gold Nanoparticles and Gelatin Nanofibers for Bone Regeneration: In Vitro and in Vivo Study. *Sci. Rep.* **2021**, *11*, 13877. [[CrossRef](#)]
12. García-Lamas, L.; Peña, J.; Roman, J.; Cabañas, V.; Bravo-Giménez, B.; Jiménez-Díaz, V.; Sánchez-Salcedo, S.; Jiménez-Holguín, J.; Abella, M.; Desco, M.; et al. In Vivo Behavior in Rabbit Radius Bone Defect of Scaffolds Based on Nanocarbonate Hydroxyapatite. *J. Biomed. Mater. Res. Part B Appl. Biomater.* **2024**, *112*, e35391. [[CrossRef](#)]
13. Spence, G.; Patel, N.; Brooks, R.; Rushton, N. Carbonate Substituted Hydroxyapatite: Resorption by Osteoclasts Modifies the Osteoblastic Response. *J. Biomed. Mater. Res. Part A* **2009**, *90*, 217–224. [[CrossRef](#)]
14. Dorozhkin, S.V.; Epple, M. Biological and Medical Significance of Calcium Phosphates. *Angew. Chemie Int. Ed.* **2002**, *41*, 3130–3146. [[CrossRef](#)]
15. Yotsova, R.; Peev, S. Biological Properties and Medical Applications of Carbonate Apatite: A Systematic Review. *Pharmaceutics* **2024**, *16*, 291. [[CrossRef](#)]
16. Peña, J.; Román, J.; Victoria Cabañas, M.; Vallet-Regí, M. An Alternative Technique to Shape Scaffolds with Hierarchical Porosity at Physiological Temperature. *Acta Biomater.* **2010**, *6*, 1288–1296. [[CrossRef](#)] [[PubMed](#)]
17. García, A.; Cabañas, M.V.; Peña, J.; Sánchez-Salcedo, S. Design of 3D Scaffolds for Hard Tissue Engineering From Apatites to Silicon Mesoporous Materials. *Pharmaceutics* **2021**, *13*, 1981. [[CrossRef](#)]
18. Cabañas, M.V.; Peña, J.; Román, J.; Vallet-Regí, M. Tailoring Vancomycin Release from β -TCP/Agarose Scaffolds. *Eur. J. Pharm. Sci.* **2009**, *37*, 249–256. [[CrossRef](#)]
19. Cabañas, M.V.; Peña, J.; Román, J.; Ramírez-Santillán, C.; Matesanz, M.C.; Feito, M.J.; Portolés, M.T.; Vallet-Regí, M. Design of Tunable Protein-Releasing Nanoapatite/Hydrogel Scaffolds for Hard Tissue Engineering. *Mater. Chem. Phys.* **2014**, *144*, 409–417. [[CrossRef](#)]
20. Paris, J.L.; Román, J.; Manzano, M.; Cabañas, M.V.; Vallet-Regí, M. Tuning Dual-Drug Release from Composite Scaffolds for Bone Regeneration. *Int. J. Pharm.* **2015**, *486*, 30–37. [[CrossRef](#)] [[PubMed](#)]
21. García-Honduvilla, N.; Coca, A.; Ortega, M.A.; Trejo, C.; Román, J.; Peña, J.; Cabañas, M.V.; Vallet Regí, M.; Buján, J. Improved Connective Integration of a Degradable 3D-Nano-Apatite/Agarose Scaffold Subcutaneously Implanted in a Rat Model. *J. Biomater. Appl.* **2018**, *33*, 741–752. [[CrossRef](#)] [[PubMed](#)]

22. Paris, J.L.; Lafuente-Gómez, N.; Cabañas, M.V.; Román, J.; Peña, J.; Vallet-Regí, M. Fabrication of a Nanoparticle-Containing 3D Porous Bone Scaffold with Proangiogenic and Antibacterial Properties. *Acta Biomater.* **2019**, *86*, 441–449. [[CrossRef](#)]
23. Almubarak, S.; Nethercott, H.; Freeberg, M.; Beaudon, C.; Jha, A.; Jackson, W.; Marcucio, R.; Miclau, T.; Healy, K.; Bahney, C. Tissue Engineering Strategies for Promoting Vascularized Bone Regeneration. *Bone* **2016**, *83*, 197–209. [[CrossRef](#)]
24. Costa, P.F. Bone Tissue Engineering Drug Delivery. *Curr. Mol. Biol. Rep.* **2015**, *1*, 87–93. [[CrossRef](#)]
25. Lee, C.-S.; Singh, R.K.; Hwang, H.S.; Lee, N.-H.; Kurian, A.G.; Lee, J.-H.; Kim, H.S.; Lee, M.; Kim, H.-W. Materials-Based Nanotherapeutics for Injured and Diseased Bone. *Prog. Mater. Sci.* **2023**, *135*, 101087. [[CrossRef](#)]
26. Hasan, A.; Morshed, M.; Memic, A.; Hassan, S.; Webster, T.J.; Marei, H.E.S. Nanoparticles in Tissue Engineering: Applications, Challenges and Prospects. *Int. J. Nanomed.* **2018**, *13*, 5637–5655. [[CrossRef](#)]
27. Kunrath, M.F.; Shah, F.A.; Dahlin, C. Bench-to-Bedside: Feasibility of Nano-Engineered and Drug-Delivery Biomaterials for Bone-Anchored Implants and Periodontal Applications. *Mater. Today Bio* **2023**, *18*, 100540. [[CrossRef](#)]
28. Saha, K.; Agasti, S.S.; Kim, C.; Li, X.; Rotello, V.M. Gold Nanoparticles in Chemical and Biological Sensing. *Chem. Rev.* **2012**, *112*, 2739–2779. [[CrossRef](#)]
29. Yadid, M.; Feiner, R.; Dvir, T. Gold Nanoparticle-Integrated Scaffolds for Tissue Engineering and Regenerative Medicine. *Nano Lett.* **2019**, *19*, 2198–2206. [[CrossRef](#)] [[PubMed](#)]
30. Gupta, A.; Singh, S. Multimodal Potentials of Gold Nanoparticles for Bone Tissue Engineering and Regenerative Medicine: Avenues and Prospects. *Small* **2022**, *18*, e2201462. [[CrossRef](#)] [[PubMed](#)]
31. Yang, D.H.; Nah, H.; Lee, D.; Min, S.J.; Park, S.; An, S.H.; Wang, J.; He, H.; Choi, K.S.; Ko, W.K.; et al. A Review on Gold Nanoparticles as an Innovative Therapeutic Cue in Bone Tissue Engineering: Prospects and Future Clinical Applications. *Mater. Today Bio* **2024**, *26*, 101016. [[CrossRef](#)]
32. Liang, H.; Xu, X.; Feng, X.; Ma, L.; Deng, X.; Wu, S.; Liu, X.; Yang, C. Gold Nanoparticles-Loaded Hydroxyapatite Composites Guide Osteogenic Differentiation of Human Mesenchymal Stem Cells through Wnt/ β -Catenin Signaling Pathway. *Int. J. Nanomed.* **2019**, *14*, 6151–6163. [[CrossRef](#)] [[PubMed](#)]
33. Li, H.; Pan, S.; Xia, P.; Chang, Y.; Fu, C.; Kong, W.; Yu, Z.; Wang, K.; Yang, X.; Qi, Z. Advances in the Application of Gold Nanoparticles in Bone Tissue Engineering. *J. Biol. Eng.* **2020**, *14*, 14. [[CrossRef](#)]
34. Shi, Y.; Han, X.; Pan, S.; Wu, Y.; Jiang, Y.; Lin, J.; Chen, Y.; Jin, H. Gold Nanomaterials and Bone/Cartilage Tissue Engineering: Biomedical Applications and Molecular Mechanisms. *Front. Chem.* **2021**, *9*, 724188. [[CrossRef](#)]
35. Zhang, Y.; Wang, P.; Mao, H.; Zhang, Y.; Zheng, L.; Yu, P.; Guo, Z.; Li, L.; Jiang, Q. PEGylated Gold Nanoparticles Promote Osteogenic Differentiation in in Vitro and in Vivo Systems. *Mater. Des.* **2021**, *197*, 109231. [[CrossRef](#)]
36. Niu, C.; Yuan, K.; Ma, R.; Gao, L.; Jiang, W.; Hu, X.; Lin, W.; Zhang, X.; Huang, Z. Gold Nanoparticles Promote Osteogenic Differentiation of Human Periodontal Ligament Stem Cells via the P38 MAPK Signaling Pathway. *Mol. Med. Rep.* **2017**, *16*, 4879–4886. [[CrossRef](#)]
37. Choi, S.Y.; Song, M.S.; Ryu, P.D.; Lam, A.T.N.; Joo, S.W.; Lee, S.Y. Gold Nanoparticles Promote Osteogenic Differentiation in Human Adipose-Derived Mesenchymal Stem Cells through the Wnt/ β -Catenin Signaling Pathway. *Int. J. Nanomed.* **2015**, *10*, 4383–4392.
38. Zhang, D.; Liu, D.; Zhang, J.; Fong, C.; Yang, M. Gold Nanoparticles Stimulate Differentiation and Mineralization of Primary Osteoblasts through the ERK/MAPK Signaling Pathway. *Mater. Sci. Eng. C* **2014**, *42*, 70–77. [[CrossRef](#)]
39. Yi, C.; Liu, D.; Fong, C.-C.; Zhang, J.; Gold, M.Y. Nanoparticles Promote Osteogenic Differentiation of Mesenchymal Stem Cells through P38 MAPK Pathway. *ACS Nano* **2010**, *4*, 6439–6448. [[CrossRef](#)] [[PubMed](#)]
40. Heo, D.N.; Ko, W.K.; Bae, M.S.; Lee, J.B.; Lee, D.W.; Byun, W.; Lee, C.H.; Kim, E.C.; Jung, B.Y.; Kwon, I.K. Enhanced Bone Regeneration with a Gold Nanoparticle-Hydrogel Complex. *J. Mater. Chem. B* **2014**, *2*, 1584–1593. [[CrossRef](#)] [[PubMed](#)]
41. Kimling, J.; Maier, M.; Okenve, B.; Kotaidis, V.; Ballot, H.; Plech, A. Turkevich Method for Gold Nanoparticle Synthesis Revisited. *J. Phys. Chem. B* **2006**, *110*, 15700–15707. [[CrossRef](#)] [[PubMed](#)]
42. Padilla, S.; Izquierdo-Barba, I.; Vallet-Regí, M. High Specific Surface Area in Nanometric Carbonated Hydroxyapatite. *Chem. Mater.* **2008**, *20*, 5942–5944. [[CrossRef](#)]
43. Román, J.; Cabañas, M.V.; Peña, J.; Vallet-Regí, M. Control of the Pore Architecture in Three-Dimensional Hydroxyapatite-Reinforced Hydrogel Scaffolds. *Sci. Technol. Adv. Mater.* **2011**, *12*, 045003. [[CrossRef](#)]
44. Ma, X.; Sun, J.; Zhong, L.; Wang, Y.; Huang, Q.; Liu, X.; Jin, S.; Zhang, J.; Liang, X.J. Evaluation of Turning-Sized Gold Nanoparticles on Cellular Adhesion by Golgi Disruption in Vitro and in Vivo. *Nano Lett.* **2019**, *19*, 8476–8487. [[CrossRef](#)]
45. Zhang, Z.; Jia, J.; Lai, Y.; Ma, Y.; Weng, J.; Sun, L. Conjugating Folic Acid to Gold Nanoparticles through Glutathione for Targeting and Detecting Cancer Cells. *Bioorg. Med. Chem.* **2010**, *18*, 5528–5534. [[CrossRef](#)]
46. Zabetakis, K.; Ghann, W.E.; Kumar, S.; Daniel, M.C. Effect of High Gold Salt Concentrations on the Size and Polydispersity of Gold Nanoparticles Prepared by an Extended Turkevich-Frens Method. *Gold Bull.* **2012**, *45*, 203–211. [[CrossRef](#)]

47. Tentor, F.R.; de Oliveira, J.H.; Scariot, D.B.; Lazarin-Bidóia, D.; Bonafé, E.G.; Nakamura, C.V.; Venter, S.A.S.; Monteiro, J.P.; Muniz, E.C.; Martins, A.F. Scaffolds Based on Chitosan/Pectin Thermosensitive Hydrogels Containing Gold Nanoparticles. *Int. J. Biol. Macromol.* **2017**, *102*, 1186–1194. [[CrossRef](#)]
48. Quarles, L.D.; Yohay, D.A.; Lever, L.W.; Caton, R.; Wenstrup, R.J. Distinct Proliferative and Differentiated Stages of Murine MC3T3-E1 Cells in Culture: An in Vitro Model of Osteoblast Development. *J. Bone Miner. Res.* **1992**, *7*, 683–692. [[CrossRef](#)]
49. Hong, D.; Chen, H.X.; Yu, H.Q.; Liang, Y.; Wang, C.; Lian, Q.Q.; Deng, H.T.; Ge, R.S. Morphological and Proteomic Analysis of Early Stage of Osteoblast Differentiation in Osteoblastic Progenitor Cells. *Exp. Cell Res.* **2010**, *316*, 2291–2300. [[CrossRef](#)]
50. Yoon, H.; Park, S.G.; Shin, H.R.; Kim, K.T.; Cho, Y.D.; Moon, J.I.; Kim, W.J.; Ryoo, H.M. Unraveling the Dynamics of Osteoblast Differentiation in MC3T3-E1 Cells: Transcriptomic Insights into Matrix Mineralization and Cell Proliferation. *Bone* **2025**, *194*, 117442. [[CrossRef](#)]
51. Gumbiner, B.M. Cell Adhesion: The Molecular Basis of Tissue Architecture and Morphogenesis. *Cell* **1996**, *84*, 345–357. [[CrossRef](#)]
52. Mathieu, P.S.; Lobo, E.G. Cytoskeletal and Focal Adhesion Influences on Mesenchymal Stem Cell Shape, Mechanical Properties, and Differentiation down Osteogenic, Adipogenic, and Chondrogenic Pathways. *Tissue Eng. Part B Rev.* **2012**, *18*, 436–444. [[CrossRef](#)]
53. Kalaskar, D.M.; Demoustier-Champagne, S.; Dupont-Gillain, C.C. Interaction of Preosteoblasts with Surface-Immobilized Collagen-Based Nanotubes. *Colloids Surfaces B Biointerfaces* **2013**, *111*, 134–141. [[CrossRef](#)] [[PubMed](#)]
54. Chen, X.; Fu, X.; Shi, J.g.; Wang, H. Regulation of the Osteogenesis of Pre-Osteoblasts by Spatial Arrangement of Electrospun Nanofibers in Two- and Three-Dimensional Environments. *Nanomed. Nanotechnol. Biol. Med.* **2013**, *9*, 1283–1292. [[CrossRef](#)]
55. McBeath, R.; Pirone, D.M.; Nelson, C.M.; Bhadriraju, K.; Chen, C.S. Cell Shape, Cytoskeletal Tension, and RhoA Regulate Stem Cell Lineage Commitment. *Dev. Cell* **2004**, *6*, 483–495. [[CrossRef](#)]
56. Wu, M.-C.; Yu, H.W.; Chen, Y.-Q.; Ou, M.-H.; Serrano, R.; Huang, G.-L.; Wang, Y.-K.; Lin, K.-H.; Fan, Y.-J.; Wu, C.-C.; et al. Early Committed Polarization of Intracellular Tension in Response to Cell Shape Determines the Osteogenic Differentiation of Mesenchymal Stromal Cells. *Acta Biomater.* **2023**, *163*, 287–301. [[CrossRef](#)]
57. Addison, W.N.; Nelea, V.; Chicatun, F.; Chien, Y.C.; Tran-Khanh, N.; Buschmann, M.D.; Nazhat, S.N.; Kaartinen, M.T.; Vali, H.; Tecklenburg, M.M.; et al. Extracellular Matrix Mineralization in Murine MC3T3-E1 Osteoblast Cultures: An Ultrastructural, Compositional and Comparative Analysis with Mouse Bone. *Bone* **2015**, *71*, 244–256. [[CrossRef](#)]
58. Casarrubios, L.; Cicuéndez, M.; Polo-Montalvo, A.; Feito, M.J.; Martínez-del-Pozo, Á.; Arcos, D.; Duarte, I.F.; Portolés, M.T. Metabolomic Characterization of MC3T3-E1pre-Osteoblast Differentiation Induced by Ipriflavone-Loaded Mesoporous Nanospheres. *Biomater. Adv.* **2025**, *166*, 214085. [[CrossRef](#)]
59. Zhang, Y.; Kong, N.; Zhang, Y.; Yang, W.; Yan, F. Size-Dependent Effects of Gold Nanoparticles on Osteogenic Differentiation of Human Periodontal Ligament Progenitor Cells. *Theranostics* **2017**, *7*, 1214–1224. [[CrossRef](#)] [[PubMed](#)]
60. Wang, Q.; Liu, J.; Yang, X.; Zhou, H.; Li, Y. Gold Nanoparticles Enhance Proliferation and Osteogenic Differentiation of Periodontal Ligament Stem Cells by PINK1-Mediated Mitophagy. *Arch. Oral Biol.* **2023**, *150*, 105692. [[CrossRef](#)] [[PubMed](#)]
61. Gao, W.; Liang, C.; Zhao, K.; Hou, M.; Wen, Y. Multifunctional Gold Nanoparticles for Osteoporosis: Synthesis, Mechanism and Therapeutic Applications. *J. Transl. Med.* **2023**, *21*, 889. [[CrossRef](#)] [[PubMed](#)]

Disclaimer/Publisher’s Note: The statements, opinions and data contained in all publications are solely those of the individual author(s) and contributor(s) and not of MDPI and/or the editor(s). MDPI and/or the editor(s) disclaim responsibility for any injury to people or property resulting from any ideas, methods, instructions or products referred to in the content.